

KHESIN, Ye. B.

"Study of the Use and Repair of Equipment Used in the Petroleum Industry" (Opyt Ekspluatatsiy i Remonta Oborudovaniya v Neftyanoy Promyshlennosti), USSR Ministry of Petroleum, 1953

Abstracts - D 138264, 3 Jan 55

KHEGIN, Ye. S.

Dissertation defended for the degree of Candidate of Economic Sciences at the  
Institute of World Economics and International Relations

"Insurance Monopolies in the British Economy."

Vestnik Akad. Nauk, No. 4, 1963, pp 119-145

KHESIN, Yefim Samuilovich; ROTOVA, R.S., red.izd-va; MURASHOVA, V.A.,  
tekhn. red.

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1963. 67 p. (MIRA 16:7)

(Insurance) (Trusts, Industrial)

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otv. red.; PLISKINA, Ye.M., red.; ASTAF'YEVA, T.A., tekhn.  
red.

[Insurance monopolies and their role in the economy and  
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ekonomike i politike Anglii. Moskva, Izd-vo Akad. nauk SSSR,  
1963. 287 p. (MIRA 16:7)

(Great Britain--Insurance)  
(Great Britain--Trusts, Industrial)

*KHESIN, Ye.Ye.*

ARUTYUNOV, V.Ya., prof.; BERKOVICH, I.M., doktor med.nauk; BUNIN, K.V., prof.  
VELIKORETSKIY, A.N., prof.; GAMBURG, R.L., doktor med.nauk; BLASKO,  
N.M.; ZVYAGINTSEVA, S.G., doktor med.nauk; IVENSKAYA, A.M., kand.med.  
nauk; KALUGINA, A.N., kand.med.nauk; KAMINSKAYA-PAVLOVA, Z.A., prof.  
KVATER, Ye.I., prof.; KOLEN'KO, A.B., prof.; KOSSYURA, M.B., kand.  
med.nauk; KRAVETS, N.M., doktor med.nauk; KRISTMAN, V.I., kand.med.  
nauk; KRUZHKOV, V.A., dotsent; LIKHACHEV, A.G., prof.; LUKOMSKIY, I.G.,  
prof.; MASHKOVSKIY, M.D., prof.; ROZENTAL', A.S., prof.; SEMENSKIY,  
M.Ya. [deceased], prof.; TURETSKIY, M.Ya., kand.med.nauk; KHESIN,  
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red.; BONDAR', Z.A., red.; ZAKHAROVA, A.I., tekhn.red.

[Medical handbook for feldshers] Meditsinskii spravochnik dlia  
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Preparing ground for trench digging in the winter. Stroi, trubo-  
prov. 8 no.11: 33-34\*63 (MIRA 1787)

1. Rukovoditel' gruppy organizatsii stroitel'stva Gosudarstven-  
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KHESIN, YU. B.; SADOV, D. A.

Electric Transformers

Remodeling a high-voltage transformer of a rectifier, Rab, energ, 2, no. 1, 1952.

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KHESIN, YU. D.

MALEYEV, V. F. Inzh. i MORGENSENTERI, N. V. Inzh. KHESIN, Yu. D. St. Nauchn. Sovv.  
Leningradskiy filial Vsesoyuznogo nauchno-issledovatel'skogo instituta stroi-  
tel'nogo i dorozhnogo mashinostroyeniya

ZAMENA TSVETNYKH METALLOV V DETALYAKH STROITEL'NYKH I DOROZHNYKH MASHIN  
DREVESNYMI I TEKSTIL'NYMI PLASTIKAMI

page 143

SO: Collection of Annotations of Scientific Research Work on Construction,  
completed in 1950,  
Moscow, 1951

SOV/137-58-12-25354  
Translation from: Referativnyy zhurnal. Metallurgiya, 1958, Nr 12, p 181 (USSR)

AUTHORS: Moroz, L. S., Khesin, Yu. D., Mingin, T. E., Chernetsov, V. I.

TITLE: The Strength of Titanium (Prochnost' titana)

PERIODICAL: V sb.: Metallurgiya. Moscow-Leningrad, AN SSSR, 1957, pp 172-193

ABSTRACT: An investigation was made of the effect of low temperatures, rate and length of loading time, notching, and other external factors on the modulus of rupture of industrial Ti smelted in an electric-arc vacuum furnace. The authors discovered a sharp difference in sensitivity to notching (SN) in metals of separate smeltings which was determined by the ratio between the specific deformation work of impact stretching of smooth specimens and the  $a_k$  of notched Mesnager-type specimens. Ti which has a high SN is also sensitive to the state of the surface in notched specimens. The maximum H content of ~ 0.007 - 0.008% with which Ti retains a tolerable SN, but this figure may vary depending upon O and N content. The intensity of the effect of H on the  $a_k$  is determined by the size and type of TiH precipitation which depends upon the cooling rate from the temperature of > 400°C. Static

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The Strength of Titanium

bending tests of notched specimens showed that the magnitude of the bending deflection and the deformation work up to the appearance of the first crack, as well as the work of propagation of the crack through the entire section of the specimen at room temperature, are less in Ti than in SKhL-4 steel. In dynamic testing Ti with 0.0007% H exhibits no cold-brittleness whatever, but when affected by impurities, in particular by H, it becomes cold-brittle. An increase in H content to 0.0125% decreases  $\psi$  by 75% at  $-196^{\circ}\text{C}$ . The authors advance a hypothesis to explain the physical nature of H-brittleness of Ti by the low Sot of favorably oriented hydride inclusions. It was discovered that the strain rate has no effect on the ductility of smooth specimens of Ti enriched with H.

G. T.

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*Kherson, Yu. D.*

S2(1)	DATE : 2000 SEPTEMBER	207/320		
<p>(Top) 2) [Redacted] Institute, 1958, 85 p., 5,000 copies printed. Bulgaria, State Scientific Committee of Technical Sciences, No. 70-A-1958.</p> <p>(Bottom) 2) [Redacted] Institute, 1958, 85 p., 5,000 copies printed.</p> <p>NOTE: This note is intended for metallurgists and metallographic engineers.</p> <p>CONTENTS: This is the second volume of collected scientific papers dealing with various problems in physical metallurgy, particularly in mechanical metallurgy. The main attention is given to problems of alloying, effects of tempering on mechanical properties, use of certain elements in alloys, effects of temperature on mechanical properties, use of certain elements to investigate certain phases, isothermal treatment, strength of certain alloying elements on brittle strength, effect of certain elements on the properties of an aluminum alloy, etc. The articles are arranged mainly in two sections: types of alloys, those used with aluminum alloys.</p>				
		4/5		
<p>CONTENTS: I. Structure and Properties of Alloys. II. Structure and Properties of Specialized Alloys. III. Structure and Properties of Some Two-phase Alloys of Aluminum.</p> <p>CONTENTS: Contributions (in part). The note relates for beta brittleness to the composition of certain of the approximately 9000 steel grains which is the composition from the point of view to the first research. The heterogeneity of the distribution of beta-homogenizing elements and impurities in the grain boundaries of the alpha phase and the boundary areas leads to the formation of beta brittleness. (2) Quantitative analysis of beta brittleness in two-phase materials (a) conducting plastic deformations of the same than during a determination of softening and hardening of the two-phase regions, and (b) carrying out heat treatment. In the first case, the temperature of the two-phase regions and (c) the temperature of the alpha phase along the grain boundaries of the structure. (3) Quantitative X-ray analysis of the content of phases in the two-phase regions. (4) Analysis of the effect of the presence of phases in the ratio of the alpha to the beta phase may be 1:1. A further increase in the beta-phase content merely causes a drop in plasticity without any appreciable grain size increase.</p>				
<p>ATTACHMENT: Library of Congress S2/320</p>				

K HESIN Yu. D.

**Математические проблемы прочности твердого тела в нормальном состоянии (Some Problems in the Strength of Solids). Collection of Articles. Moscow, Izd-vo Akad. Nauk SSSR, 1959. 386 p. Printed slip inserted.**

M., of Publishing House, V. I. Aver'yanov, Tech. Ed.; R. S. Petrovsky, Academician; G. V. Kurovskii, Academy of Sciences USSR; Institute of Applied Physics, USSR Academy of Sciences, Moscow, USSR; Academy of Sciences, USSR, Moscow, 20. T. S. N. Zhdanov, Corresponding Member, USSR Academy of Sciences; Professor Y. P. Vinogradov, Doctor of Physical and Mathematical Sciences, Professor; M. A. Klinov, Doctor of Physical and Mathematical Sciences; V. A. Stoyanov, Doctor of Technical Sciences; D. N. Friedman, Doctor of Technical Sciences; Professor; B. N. Terterov, Candidate of Technical Sciences (Deputy Rapp. Sec.).

**PURPOSE:** This book is intended for construction engineers, technologists, metallurgists, physicists, and other persons interested in the strength of materials.

**CONTENTS:** This collection of articles was compiled by the Ordzhonikidze Scientific-Research Institute of Mathematics and the Physical-Mechanical Institute Akademii Nauk (Institute of Applied Physics, Institute of Technology, Member of the Ukrainian Academy of Sciences, founder and head of the Odessa Hydrocarbon Institute), in commemoration of the 50th birthday of Nikolayevich Ordzhonikidze, Member of the Ukrainian Academy of Sciences, founder of the Institute of Applied Physics, Academy of Sciences, USSR, and founder of the Pauli Test Methodology methodology. The collection includes 16 articles from the Leningradskiy politekhnicheskiy institut (Leningrad Polytechnic Institute), recipient of the Stalin Prize (1953), the Order of the Red Banner of Labor (1953) and the Order of Lenin (1953). The articles deal with the strength of materials, phenomena of fatigue, plasticity, temperature brittleness, hydrogen embrittlement, cold brittleness, influence of deformation speed on the mechanical properties of materials, fatigue of metals, and general problems of the strength, plasticity, and mechanical properties of materials. Numerous personalities are mentioned in the introductory profile of Professor Davidov. References are given at the end of each article.

Journal, I.S., and T.D. Shelekh. Investigation of the Hydrogen Embrittlement of Two-Phase Transition Alloy. 260

Dmitriev, Yu. M., and G.P. Shmel'kin. Hydrogen Embrittlement of Steel and the Influence of Mechanical Testing Conditions on Its Coarseness. 152

Ashurbayev, Yu. M., V.B. Fedorov, and S.M. Pal'mov. Institute for Metal Research, Ural Branch, Academy of Sciences, USSR. Strength of Structures of Austenitic Grain Boundaries and the Temper Brittleness of Structural Steel. 165

Savchenko, M.V., and T.A. Romanova (Institut Metallurgii Akademii Nauk SSSR, Moscow). Metallurgical Institute, Academy of Sciences USSR, Moscow. The Coarseness of the Degree of Purity on Cold Brittlezaus and Other Properties of Chromium. 172

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Khokhlov, Yu. G. (Industrialnyy Institut Metallovedeniya, Sverdlovsk). Industrial Institute (Sverdlovsk, Kirovav), Kirovav. Effect of the Cooling Rate and Some Other Factors on Impact Strength of Chromium-Alumina Steel. 187

Savchenko, Yu. M. (Deceased), I.I. Naletov, and A.Y. Isichenko. Influence of the Scale Factor During Plastic Deformation and Structure of Steels of Various Strength. 194

Vil'man, Z.F., and V.A. Shestopalov (Institute of Applied Physics, Academy of Sciences, USSR, Gorkiy). Influence of Deformation Rate on the Formation of Metal at Impact Speeds of  $10 - 10^4$ /sec. 207

Klinov, M.A. (Institute of Applied Physics, Academy of Sciences, USSR, Gorkiy). Role of Compressibility in the Dynamic Deformations of Plastics. 222

Savchenko, Yu. M., and V.I. Romanov. Influence of a High Deformation Rate on the Mechanical Properties of Steel Alloy Type V-55 After Various Degrees of Aging. 230

Gulyuk, S.Y., and Yu.N. Volobuev. Brittleness (Institute of Mechanical Engineering, Academy of Sciences, USSR, Moscow). Resistance to Plastic Deformation During Repeated Stress Under Low-Temperature Conditions. 238

Gulyuk, S.Y., and V.P. Solntsev. Physical Nature of Metal Brittle. 246

Khokhlov, Yu. M., and B.M. Savrin (FizTekhnika - Central Scientific Research Institute of Technology and Mechanics). Fatigue Strength of Large Parts. 256

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KHESIN, Yu.D., inzh.; SHUL'KIN, S.M., kand.tekhn.nauk

Plastic and heat treatment of certain binary titanium alloys.  
Metallovedenie 2:251-265. 1958.  
(Titanium alloys--Metallography) (MIRA 13:9)

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KHESEN, Yu. D.

## PHASE I BOOK INFORMATION 807/3752

Moscow Polytechnic Shment' Stroya, No. 3 (Periodical Metallostroy Collection of Articles), No. 3, Leningrad, September, 1953. 580 p., 5200 copies printed.

Sd.: G. I. Bayrkin, Candidate of Technical Sciences; Literatury and Tech. Sci. Ed.

H. I. Kavardash.

NOTICE: This collection of articles is intended for scientists, personnel, as well as research and educational institutions and industrial plants and also for advanced students.

CONTENTS: The articles report the results of investigations of 1) the effect of various factors on the ductility of structural and heat-resistant steels and titanium alloys to bending, fatigue, and impact loading; 2) properties under various conditions of loading (long-time, short-time, cyclic, monotonic); 3) allotropic structures, and condition of alloy as related to their mechanical properties; and 4) corrosion resistance and evaluation of stainless and heat-resistant steels. The articles are accompanied by numerous tables and non-tablet references. No personnel lists are mentioned.

Khvorostov, A. S., Doctor of Technical Sciences, Professor. Review of Steel Brittleness Processes During Heating and the Effects of Allotropic Elements on Them

Polyakov, V. N., Candidate of Technical Sciences; E. E. Solyanik, Engineer. Effect of Nickel on the Resistance of Chromo-Nickel-Vanadium-Vanadium Carbide to the Brittleness of Chrome-Nickel-Vanadium-Vanadium Carbide Structural Steel

Rozov, L. S., Doctor of Technical Sciences, and V. S. Knipps, Engineer. Mechanism of Hydrogen Brittleness in Steel

Gol'dman, L. A., Doctor of Technical Sciences, Professor; B. M. Kol'shina, Engineer. Effect of Hydrogen on the Mechanical Properties of Chromo-Nickel-Vanadium-Vanadium Carbide Structural Steel

Dobrolyubov, V. P., Doctor of Technical Sciences, Professor; V. F. Dzhurba, Candidate of Chemical Sciences, and V. F. Dzhurba, Candidate of Technical Sciences. Change in Mechanical Properties of Certain Steels Under the Action of Spraying at High Temperatures and Pressures

Perov, L. S., and Yu. N. Dushkin, Engineer. Investigation of the Mechanism of Hydrogen Brittleness of Titanium and Its Alloys

Mishin, S. I., Candidate of Technical Sciences. Role of Intermediate Structure in the Heat Treatment of Medium-Alloy Constructional Steel

Obukhov, I. Ya., Radiographer. Stability of Structures and Properties of Tempered Steel

Bachinskii, A. I., Candidate of Technical Sciences. Microscopic and Metallographic Study of Quench-Harden Steel

Gerasimov, V. I., Doctor. Structure of Titanium and Its Aluminides After Impact Testing Under Asymmetric Loading

Chernihals, B. B., Candidate of Technical Sciences. Investigation of the Mechanism of Formation of Specialized and Development of the First Fatigue Crack in Heating Steel for Mechanical Properties

Rubtsov, P. O., Doctor of Technical Sciences, Professor. Some Observations on the Strength of Metals as Related to Their Microstructure

Shchegolev, S. S., Candidate of Technical Sciences. Preparation of the Stress-Strain Curves or Stress-Strain Diagrams and Relaxation of Structures for

100

Khush, Yu. A., Book Treatment of Non-Ferrous Alloys of Titanium

Khush, L. S., and Shchegolev, S. S., Analysis of Strength of Metals in

101

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S/180/60/000/01/016/027  
E193/E13518.1285  
18.8200AUTHORS: Moroz, L.S., and Khesin, Yu.D. (Leningrad)TITLE: Investigation of the Mechanism of Hydrogen Embrittlement  
of Titanium and its AlloysPERIODICAL: Izvestiya Akademii nauk SSSR, Otdeleniye tekhnicheskikh  
nauk, Metallurgiya i toplivo, 1960, Nr 1, pp 111-122 (USSR)

ABSTRACT: The object of the present investigation was to study the effect of hydrogen on the mechanical properties of α-Ti and titanium alloys of the β and α+β type (the constitution diagram of the Ti-H system is reproduced in Fig 1; wt-%, top scale, at-%, bottom scale). The experimental materials comprised: technical purity Ti (U.T.S. = 55 kg/mm<sup>2</sup> at room temperature); a two-phase, Ti-base alloy containing 2% Mn, 1.3% Fe, 0.8% Cr, 1.2% Mo and 1.2% V; a two-phase, Ti-base alloy containing 5% Al, 3% Mo and 3% V; and a β-alloy, containing 15% Mo. After hot working, all these alloys were finely-crystalline with the average grain size of 0.04 to 0.07 mm; after a vacuum treatment (15 h at 700°C) their hydrogen content was less than 0.003%. Two methods were used to introduce hydrogen into the test pieces that were to be employed in the subsequent tests: the

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the heading of the table). The effect of hydrogen on the mechanical properties of the technical purity titanium, annealed at 650 °C, is illustrated by data given in Table 2 under the following headings: H<sub>2</sub> content, wt-%; σ<sub>s</sub> (yield point, kg/mm<sup>2</sup>); ψ (reduction of area, %); a<sub>k</sub> (impact strength, kg/mm<sup>2</sup>). It will be seen that whereas neither the yield point nor ductility (as indicated by ψ) of the specimens were affected by increasing hydrogen concentration, the impact strength, determined on notched bars, failed catastrophically. This effect is a direct consequence of the nature of the Ti-H constitution diagram (Fig 1). Solubility of H in α-Ti varies from 0.18 at 300 °C to 0.002 wt-% at 100 °C; after slow cooling from temperatures above 300 °C, hydrogen is present in titanium in the form of fully precipitated titanium hydride platelets (see the photomicrograph, Fig 2); when titanium, containing less than 0.18 wt-% H<sub>2</sub>, is heated to 300 °C, hydrides dissociate completely and a solid solution of H in Ti is formed. On quenching from this or a higher temperature, a super-saturated, precipitation-hardenable,

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solid solution will be obtained. Thus, a titanium specimen with 0.03% H, quenched from 500 °C, had an impact strength of 7 kgm/cm<sup>2</sup>; after ageing at 200 °C its impact strength decreased to 1 kgm/cm<sup>2</sup>. Similar results could be obtained by prolonged room temperature ageing; this is illustrated by data, given in Table 3, which shows values of  $a_k$  of the H-bearing Ti specimen after quenching from 500 °C, and after 1, 10 and 100 days ageing at room temperature. Electron-microscope study of the ageing process confirmed the hypothesis that, in this case, embrittlement during ageing is associated with the precipitation and coalescence of titanium hydrides; this is illustrated clearly by the photomicrographs (X 2350) reproduced in Fig 3 (a - the microstructure of an H-bearing, Ti specimen in the quenched condition, b - the same microsection after 7 days' ageing at room temperature) which show the increased proportion of the hydrides as well as the grain-boundary broadening in the aged material. Regarding the mechanism of the embrittling,

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effect of hydrides, the authors base their considerations on the experimental data reproduced in Tables 4 and 5. The effect of the rate of deformation on ductility of annealed, H-bearing,  $\alpha$ -Ti is illustrated in Table 4, which shows: H<sub>2</sub> content, wt-%; elongation ( $\delta$ , %) and reduction of area ( $\psi$ , %) for specimens, tested at the rates of strain of: (I) 2 mm/min, and (II) 2.10<sup>5</sup> mm/min. The effect of the test temperature on the ductility of the same material is illustrated in Table 5, showing: H<sub>2</sub> content, wt-%,  $\delta$ , and  $\psi$  determined at +20, -20 and -60 °C; (the specimen with 0.03% H tested at -60 °C failed in a brittle manner). It can be inferred from data given in Tables 2, 4 and 5 that brittleness due to hydrogen is not revealed by standard tensile tests, conducted on cylindrical specimens, and only becomes evident in the presence of a notch, at high rates of strain, or at low temperatures. These facts can be interpreted in one way only: titanium hydrides, while possessing some ductility, have low resistance to rupture; if the normal tensile stress in titanium is lower than the rupture strength of

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the hydrides, the effect of hydrogen will not become apparent; if the normal stress is raised above that critical value (by introduction of a notch, increasing the strain rate, or lowering the temperature), cracks are formed in the hydrides which reduce the strength of the metal to a level depending on the proportion of hydride platelets present and on their size, since these factors determine the number and dimensions of the cracks. This is illustrated by data reproduced in Fig 4, where the true tensile strength ( $S_k$ , kg/mm<sup>2</sup>) of H-bearing titanium at -196 °C is plotted against the quantity and dimensions of the precipitated hydrides, points a, b, c, and d relating to: (a) specimen quenched from 500 °C (low hydride concentration); (b) specimen quenched and aged for 2 h at 100 °C (medium concentration of hydrides of small size); (c) specimen annealed at 400 °C (high concentration of coarse hydride particles). The propagation of cracks in hydrogen-embrittled titanium is assisted by the internal tensile stresses, present at the

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edges of the hydride platelets owing to their higher (in comparison with Ti) specific volume. Oxygen, nitrogen, and carbon additions increase the sensitivity of titanium to hydrogen embrittlement, since they promote propagation of cracks; the effect of aluminium is beneficial since this metal increases solubility of hydrogen in titanium. The effect of hydrogen on the mechanical properties of a  $\beta$ -type, 15% Mo-Ti alloy was studied next. The results are reproduced in Table 6, showing: condition of the alloy (degassed; hydrogen-impregnated by electrolytic treatment - 3 h at 0.2 Amp/cm<sup>2</sup>); U.T.S. ( $\sigma_B$ , kg/mm<sup>2</sup>); yield point ( $\sigma_S$ , kg/mm<sup>2</sup>);  $\delta$ .%;  $\psi$ , %. It will be seen that none of the investigated properties were affected by the presence of hydrogen. The results of experiments on specimens with higher content of hydrogen (introduced by high-temperature diffusion), quenched from 750 °C, are given in Fig 5, where  $\psi$  of specimens tested at the rates of strain of 2 and 200 mm/min (crosses and circles, respectively) is plotted against the hydrogen content (%). ✓

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It will be seen that as long as hydrogen is in the solution, it does not affect the ductility of the  $\beta$ -phase; precipitation of hydrides in the  $\beta$ -phase causes the metal to fail in a brittle manner, this effect being attributed to notch-sensitivity of the  $\beta$ -phase. The hydrogen embrittlement of the  $\alpha+\beta$  alloys is next discussed. Two alloys of this type, containing 20 and 50% of the  $\beta$ -phase, were investigated. Their mechanical properties ( $\sigma_s$ ,  $\delta$ , and  $\psi$ ), are given in Table 7, the figures in the first and second sub-columns for each property relating to the hydrogen-free specimens and to specimens subjected to 24 h electrolytic hydrogenation treatment. It will be seen that, whereas the yield point was not affected by the presence of hydrogen, the ductility of the alloy ( $\delta$ ,  $\psi$ ) decreased sharply. It was observed, also, that fracture of the hydrogen-bearing specimens started at the surface, the first cracks appearing already in the elastic deformation range (see Fig 6). The effect of the

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variation of the content of hydrogen, introduced by high-temperature diffusion, is illustrated by data given in Table 8 under the following headings: H<sub>2</sub> content, wt-%;  $\psi$ , %, of the alloy containing 20 and 50% of the  $\beta$ -phase. (A specimen of the alloy, containing 20% of the  $\beta$ -phase and 0.1% H<sub>2</sub>, failed in the brittle manner). These results showed that the embrittling effect of hydrogen was more pronounced in the alloy with a lower content of the  $\beta$ -phase. The effect of the deformation rate is illustrated in Figs 7 and 8. In Fig 7a,  $\psi$  is plotted against the rate of strain (V, mm/min) for an alloy containing 20% of the  $\beta$ -phase, curves 1 and 2 relating to specimens before and after the electrolytic hydrogenation treatment, respectively; the corresponding curves for the alloy containing 50% of the  $\beta$ -phase are plotted in Fig 7b. In Fig 8a,  $\psi$  is plotted against V for the alloy containing 50% of the  $\beta$ -phase, curves 1, 2 and 3 relating to specimens with 0.025, 0.050 and 0.1% of hydrogen (introduced by high-temperature diffusion)

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treatment) respectively; the corresponding curves (1 and 3) for the alloy containing 20% of the  $\beta$ -phase are plotted in Fig 8b. In this case, too, the proportion of the  $\beta$ -phase determined the behaviour of the alloys. The ductility of specimens containing hydrogen, introduced electrolytically, increased with increasing V, approaching the ductility of hydrogen-free material at  $V = 200$  mm/min, this restoration of ductility with increasing V being less pronounced in the alloy with 50% of the  $\beta$ -phase. In the case of specimens containing hydrogen introduced by the high-temperature diffusion treatment, the restoration of ductility with increasing V was slow in specimens containing 50% of the  $\beta$ -phase, and did not occur at all in specimens containing 20% of the  $\beta$ -phase and 0.1% Ti<sub>2</sub>. The effect of the constitution on the sensitivity of the  $\alpha+\beta$  alloys to hydrogen embrittlement was revealed also by the results of impact strength tests, conducted on notched, cylindrical specimens 8 mm diameter (depth of 4

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the notch 1 mm, root radius 0.55 mm). The results of these tests are given in Table 9, showing: H<sub>2</sub> content, wt-%;  $\sigma_k$ , kgm/cm<sup>2</sup>, of specimens containing 20 and 50% of the  $\beta$ -phase. However, it is pointed out that tensile test at slow rates of strain is a more sensitive method of revealing the hydrogen embrittlement of titanium alloys of the  $\alpha+\beta$  type. The difference in the behaviour of material containing hydrogen, introduced by different techniques, is attributed to the fact that hydrogen introduced electrolytically (i.e. at room temperature) can dissolve in the  $\beta$ -phase only. This was checked by X-ray diffraction analysis, carried out on a complex, Mn-bearing alloy, whose alloying elements, however, did not affect the solubility of hydrogen. The results are given in Table 10 under the following headings: constitution of the alloy (relative proportion of the  $\alpha$ - and  $\beta$ -phase); lattice parameters of the  $\alpha$ - and  $\beta$ -phases in the degassed alloy; lattice parameters of the  $\alpha$ - and  $\beta$ -phases in the alloy with

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Investigation of the Mechanics of Hydrogen Embrittlement of  
Titanium and its Alloys

electrolytically introduced hydrogen. When hydrogen is introduced by the high-temperature diffusion treatment and the alloy is subsequently heated to the quenching temperature, re-distribution of hydrogen (i.e. its diffusion into the  $\beta$ -phase) takes place, until the partial pressure of hydrogen in both phases becomes equal. In the case under consideration, the re-distribution of hydrogen between the two phases is caused by the fact that at a given temperature, the equilibrium partial pressure of hydrogen, dissolved in the  $\alpha$ -phase, is higher than that of hydrogen dissolved in the  $\beta$ -phase. This is illustrated by the diagram, reproduced in Fig 9, where the equilibrium partial pressure ( $P$ , mm Hg) is plotted against temperature ( $^{\circ}\text{C}$ ) for an alloy containing 2 at-%  $\text{H}_2$ . The non-uniform distribution of hydrogen in a two-phase alloy can be arrested by quenching. However, for a given hydrogen content in the specimen, its concentration in the  $\beta$ -phase will not be constant (as in the case of specimens with electrolytically introduced hydrogen), but ✓

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## Investigation of the Mechanics of Hydrogen Embrittlement of Titanium and its Alloys

will increase with decreasing proportion of the  $\beta$ -phase in the alloy. In addition, owing to solubility of hydrogen in the  $\alpha$ -phase decreasing with falling temperature, the formation of hydrides may occur in this phase. Since solubility of hydrogen in titanium can be increased by alloying, the alloy used in the next series of experiments, in addition to 3% Mo and 3% V (elements stabilizing the  $\beta$ -phase), contained 5% aluminium which considerably increases solubility of hydrogen in the  $\alpha$ -phase. The results are given in Table 11, showing: H<sub>2</sub> content, wt-%;  $\delta$ , and  $\psi$  for specimens tested at the rate of strain of (I) 2 mm/min and (II) 200 mm/min. It will be seen that the presence of aluminium considerably decreased the proneness of the alloy to hydrogen embrittlement. The results of mechanical tests, conducted on specimens of the same alloy containing electrolytically introduced hydrogen, are given in Table 12, showing: condition of the specimen (treated electrolytically at  $I = 0.2$  amp/cm<sup>2</sup> for 1.5 h, top line, and at  $I = 0.25$  amp/cm<sup>2</sup> for 3.0 h, bottom line);  $\psi$  of

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## Investigation of the Mechanics of Hydrogen Embrittlement of Titanium and its Alloys

specimens tested at the rate of strain of 2 mm/min (I) and 250 mm/min (II). In this case, the beneficial effect of aluminium was also apparent. Analysis of the results of the present investigation, correlated with the findings of other workers, led the present authors to several conclusions. (1) The sensitivity of single-phase titanium to hydrogen embrittlement is determined by two factors: (a) room temperature solubility of hydrogen in the given phase which determines the "safe" and "dangerous" range of hydrogen concentration; (b) ease with which the cracks, caused by the presence of hydrides, are propagated throughout the alloy; this characteristic depends on the magnitude of the work of deformation done in propagating the cracks in titanium. The "safe" hydrogen concentration in the  $\beta$ -phase is one thousand times higher than that in the  $\alpha$ -phase. On the other hand, cracks are propagated more easily in the  $\beta$ -phase, the  $\alpha$ -phase being less notch-sensitive and showing no tendency to cold shortness. It is precisely owing to the ✓

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Investigation of the Mechanics of Hydrogen Embrittlement of  
Titanium and its Alloys

ease of propagation of cracks in the  $\beta$ -phase that the fully brittle condition was observed in the  $\beta$ -alloy immediately after the appearance of the first hydride precipitates (see Fig 5), whereas in the case of specimens with the hydrogen content below the saturation point, increasing the hydrogen content had practically no effect on the mechanical properties of the alloy.

(2) The hypothesis that hydrogen embrittlement of the two-phase titanium alloys can be attributed to the effect of hydrogen on each of the phases separately, has not been confirmed by the results of the present investigation. It is more likely that in the case of two-phase alloys, it is the interphase boundaries (absent in single-phase alloys) which constitute the geometrical locus of hydrogen embrittlement. Since hydrogen embrittlement in titanium alloys is directly associated with the presence of hydrides, and since microscopic examination of alloys, whose ductility depends on the rate of deformation, has revealed no hydrides, it has to be assumed that hydrides are precipitated in these alloys at

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AUTHORS:

Moroz, L.S., Khesin, Yu.D.

68986

S/020/60/151/02/025/071  
B013/B011

TITLE:

Anomalous Growth of Metal Grains in Vacuo

PERIODICAL:

Doklady Akademii nauk SSSR, 1960, Vol 131, Nr 2, pp 306-307 (USSR)

ABSTRACT:

The authors made a special investigation of the influence exerted by the annealing of various metals in vacuo on the growth of their grains. This was done for the following reasons: In the investigation of the particular behavior of metals at high temperatures in vacuo they found a new phenomenon, namely, a faster growth of the grains than during annealing in normal air. In annealing deformed titanium in vacuo ( $1.10^{-4}$  torr) at temperatures above  $800^{\circ}$  this grain growth becomes clearly noticeable. The following materials were used for these investigations: technically pure titanium, Armco iron, M1-type copper. A fine-grained structure of these materials was obtained by deformation and annealing (grain size after annealing: 30 to  $50\mu$ ). The annealed samples of all metals were mechanically treated and afterwards cut into two halves. One half of each sample was treated in vacuo, the other half in air in usual furnaces. Figure 2 shows microphotographs of the samples after annealing in vacuo and air (for iron, copper, and titanium). The grain size of all metals investigated was considerably larger after annealing in vacuo than after annealing in air. The higher the temperature of annealing the

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more distinct is this difference. Figure 4 shows the photographs of titanium macrostructure which were taken from sections of the samples. The sample annealed in vacuo had a coarse structure all over the cross section, the sample annealed in air was fine-grained. Armco iron samples gave analogous results. In the following one of the possible explanations of the anomalous growth of metal grains in vacuo is given: Various impurities having a higher vapor pressure than the metallic solvent are removed intensely during vacuum annealing. Experiments made by J.C. Chaston (Ref 2) concerning the growth of silver grains are considered to be very interesting in this respect. In heating deformed technically pure silver in air the grain grows only in the center of the sample, whereas in the outer layers (where oxygen could diffuse into silver) a very fine grain could develop. For a precise determination of the influence exerted by the impurities removed from the metal on the growth of the grain, titanium-<sup>1</sup> and Armco iron<sup>2</sup> samples were annealed at 1200° in vacuo for five hours, subjected to cold deformation (50%) afterwards, and then annealed at 650°. After mechanical surface treatment these samples were cut into two pieces one of which was annealed in vacuo, the other in air. Temperature and duration of the second annealing process in vacuo were considerably less than the

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first time. In this case the grain size of the samples annealed in vacuo and in air did not show any difference. Finally, an argument for the decisive influence of impurities on the growth of grains is given. There are 4 figures and 8 references, 1 of which is Soviet.

PRESENTED: October 20, 1959, by G.V. Kurdyumov, Academician ✓  
SUBMITTED: October 15, 1959

Card 3/3

18.8200

39764  
S/126/62/013/006/011/018  
E021/E192

AUTHORS: Moroz, L.S., Khesin, Yu.D., and Marinets, T.K.

TITLE: Study of creep and long-term strength of iron at low temperatures

PERIODICAL: Fizika metallov i metallovedeniye, v.13, no.6, 1962,  
912-919

TEXT: The main investigations were carried out on Armco iron containing 0.1% C, 0.034% N, 0.18% O<sub>2</sub> and 0.06% Cu. The samples were tested after annealing at 930 °C. The deformation during creep was measured with an accuracy of  $4 \times 10^{-5}$  cm. The test temperatures were obtained using mixtures of dry ice in kerosene (-40 °C) and in benzene (-75 °C). With a stress of 34 kg/mm<sup>2</sup>, creep occurred at -40 and -75 °C and on the steady-state part of the curve the rate was  $10^{-2}$  to  $10^{-3}$  %/hour. At room temperature there was no steady-state creep at this stress. It is proposed that the reason for the absence of creep effects at 18 °C is due to the influence of deformation ageing of iron. The energy of activation of the process of creep fracture for low temperatures and for a stress of 39 kg/mm<sup>2</sup> was found to be 13.5 kcal/mol. f

Card 1/2

MOROZ, L.S.; KHEGIN, Yu.D.; MARINETS, T.K.

Investigating creep and the stress-rupture strength of  
iron at low temperatures. Fiz. met. i metalloved. 13 no.6:912-919  
Je '62. (MIRA 15:?)

(Iron--Testing)  
(Metals at low temperatures)

S/129/63/000/002/004/014  
E193/E383

AUTHORS: Moroz, L.S., Khesin, Yu.D. and Belova, O.S.

TITLE: Structure and mechanical properties of low-alloy titanium alloys

PERIODICAL: Metallovedeniye i termicheskaya obrabotka metallov,  
no. 2, 1963, 17 - 23

TEXT: The object of the present investigation was to determine the cause of deterioration in strength and plasticity suffered by titanium alloys of a composition near to that of the  $\alpha$ -phase when they are slowly cooled from the  $\beta$  range. The experimental materials included titanium iodide, technical-purity titanium and Ti - 4% Al alloys, containing 0.55 - 1.62% V, 0.64 - 1.36% Mo, 0.66 - 1.27% Mn or 0.71% Fe. The effect of the following treatments was studied: 1) annealing at 800  $^{\circ}$ C for 2 hours; 2) water-quenching from 1 250  $^{\circ}$ C; 3) furnace-cooling from 1 250  $^{\circ}$ C. The first series of tests comprised determination of the mechanical properties. Typical results for pure and alloyed titanium are given below.

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Alloy	Heat treat- ment	$\sigma_b'$	$\sigma_{0.2}$	$\delta$	$\Psi$	$a_k$
Titanium iodide	1	31.0	22.4	60.5	82.6	30.2
	2	33.0	23.7	44.6	80.9	25.1
	3	31.3	19.9	58.9	83.0	26.1
Ti - 4% Al- - 0.71% Fe	1	74.2	69.4	16.1	46.0	8.2
	2	81.6	73.9	16.4	43.4	9.5
	3	64.6	59.0	8.9	25.3	5.5

Key:  $\sigma_b'$  = UTS, kg/mm<sup>2</sup>;  $\sigma_{0.2}$  = 0.2% proof stress;  $\delta$  = elongation, %;  
 $\Psi$  = reduction in area, %;  $a_k$  = impact strength, kgm/cm<sup>2</sup>.

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E193/E383

Structure and ....

To determine the cause of marked differences between the effect of slow cooling on the properties of pure and alloyed Ti, the micro-structure of specimens subjected to various heat-treatments was studied, the composition of the second phase found in slowly-cooled alloys was determined and its effect on the mode of plastic deformation was studied by microscopic examination of test pieces extended to various degrees of deformation and by following the changes taking place on the surface of preliminarily polished tensile test pieces during the actual tensile test. Conclusions: 1) decreasing the rate at which Ti alloys, containing small additions of the  $\beta$ -phase stabilizing elements, are cooled from the  $\beta$  range brings about a change in the structure of the alloy grains and a decrease in the mechanical properties. 2) The structural change consists of the appearance of plate-like precipitates of the second phase, formed above 800°C, i.e. in the  $\beta \rightarrow \alpha + \beta$  transformation range. 3) The presence of these precipitates leads to nonuniform deformation; as a result, microcracks are formed in the region of localized deformation in the early stages of plastic flow and this causes a decreases in strength and plasticity of the alloy.

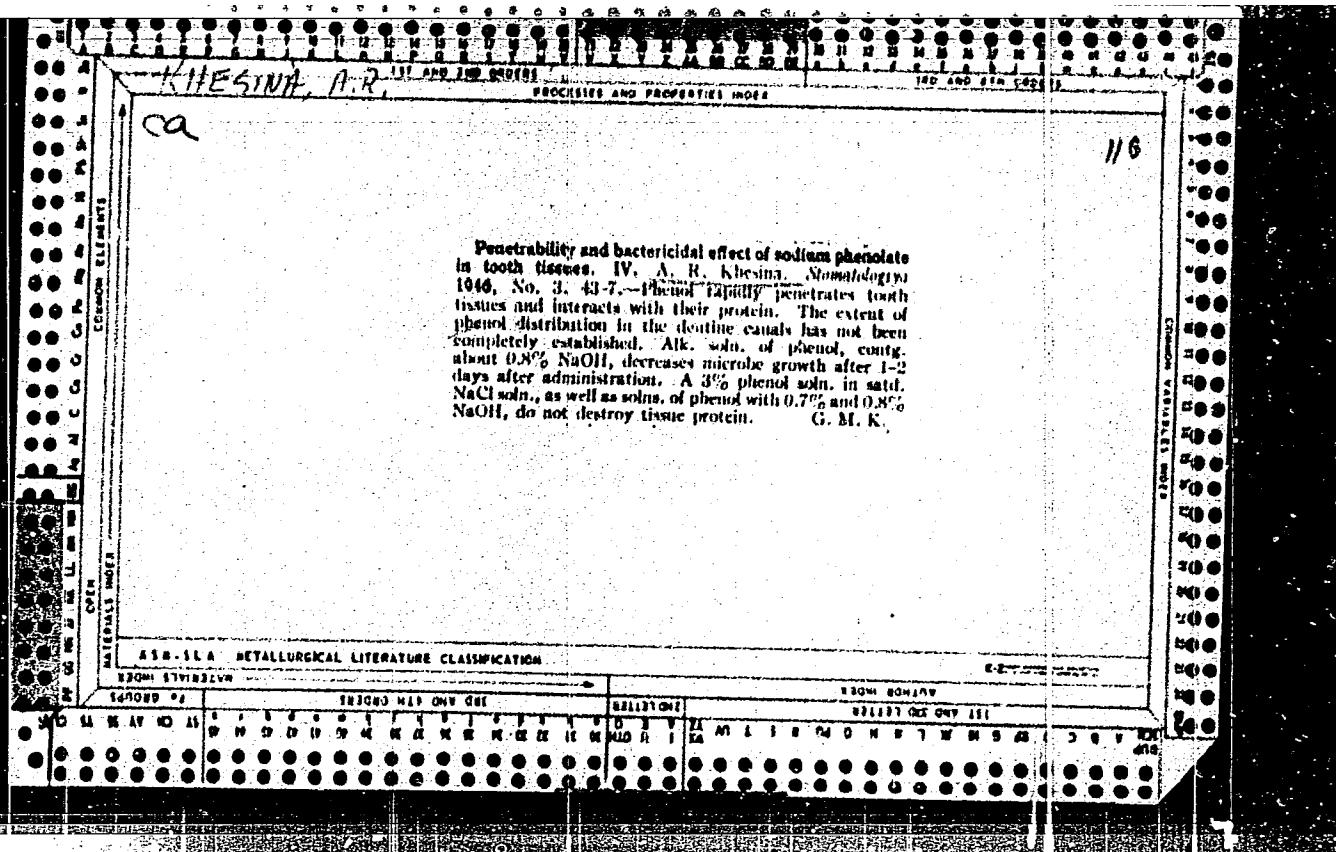
Card 3/4

Structure and ...

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4) The harmful effect of the second-phase precipitates increases with increasing distance between them which, in turn, depends on the rate of cooling of the specimens from the  $\beta$  range. 5) The results of X-ray and spectrographic analysis show that the formation of plate-like precipitates is associated with redistribution of the  $\beta$ -phase stabilizing elements; the concentration of these elements in the precipitate is so high that the  $\beta$ -phase is retained in the precipitate at room temperature. The fact that formation of second-phase precipitates occurs only in slowly-cooled specimens indicates the diffusion character of the process.  
There are 6 figures and 7 tables.

Card 4/4



KHESINA, A.R.

36460. KHESINA, A. R. I SEREBRYAKOVA, V. M.  
Profilakticheskaya Flyuorizatsiya Po Lukomskomu V Gal'Vanicheskem Tsekhe.  
Stomatologiya, 1949, No. 4, S. 28-29.

SO: Letopis' Zhurnal'nykh Statey, Vol. 49, Moskva, 1949

SHABAD, L.M.; KHESINA, A.Ya.

Spectral-luminescent determination of 3,4-benzopyrene in  
high-purity industrial paraffins. Zav. lab. 31 no.11:  
1345-1347 '65. (MIRA 19:1)

1. Institut eksperimental'noy i klinicheskoy onkologii AMN SSSR.

KHESINA, A.Ya.

Spectroscopy of certain pyrene derivatives in frozen solutions.  
Izv.AN SSSR.Ser.fiz. 24 no.5:623-626 My '60.

(MIRA 13:5)

1. Gosudarstvennyy pedagogicheskiy institut im. V.I.Lenina.  
(Pyrene---Spectra)

KHESINA, A.Ye.

Emission and absorption spectra of frozen crystalline  
solutions of certain pyrene derivatives. Opt.i spektr. 10  
no.5:607-616 My '61.

(MIRA 14:8)

(Pyrene—Spectra)

L 19470-63

EVA(b)/EWP(j)/EFF(c)/EWT(l)/EWT(m)/BDS AFFTC/ASD/IJP(c)/SSD

ACCESSION NR: AT3002193 Pa-4/Pc-4/Pr-4 RM/WH/MAY S/2941/63/001/000/0043/0051

AUTHOR: Khesina, A. Ya.TITLE: Fluorescence spectra of n-paraffin solutions of pyrenes

SOURCE: Optika i spektroskopiya; sbornik statey. v. 1: Lyuminestsentsiya. Moscow, Izd-vo AN SSSR, 1963, 43-51

TOPIC TAGS: pyrene, spectra, fluorescence, n-paraffin

ABSTRACT: An experimental method was developed to obtain the fluorescence spectra of n-paraffin solutions of some pyrene derivatives in a temperature range -196 to 0C. The purpose was to determine the effect of temperature on the quasi-line fluorescence spectra of frozen crystalline solutions of some pyrenes, at intervals of 77K. The pyrenes used were: 3,4-naphtho-6,7-benzopyrene in n-hexane; 3-methyl-4,5-ethylene-3,4,6,7-dibenzopyrene in n-octane; 3,4,5,6,7-tribenzopyrene in n-octane. It is shown that at the melting point the spectra blur into a wide band which remains the same up to room temperature. At 77K the spectra of the complex pyrenes show a quasi-linear form. "The author expresses his gratitude

Card 1/2

L 19470-63

ACCESSION NR: AT3002193

to E. V. Shpol'skiy, B. A. Arbuzov and T. N. Bolotnikova." Orig. art. his: 5  
figures, 3 formulas, and 1 table.

3

ASSOCIATION: none

SUBMITTED: 06Apr62

DATE ACQ: 19May63

ENCL: 00

SUB CODE: PH

NO REF SOV: 010

OTHER: 003

Card 2/2

KHESINA, B.G.

3(7)

PHASE I BOOK EXPLOITATION

SOV/3031

Moscow. Tsentral'nyy institut prognozov

Voprosy dolgosrochnykh prognozov (Problems in Long-Range Forecasting)  
Moscow, Gidrometeoizdat (otd.) 1958. 104 p. (Series: Its: Trudy,  
vyp. 73) 1,100 copies printed.

Sponsoring Agency: USSR. Glavnoye upravleniye gidrometeorologicheskoy  
sluzhby.

Ed.: (title page): V.M. Kurganskaya; Ed. (inside book): V.I. Tarukhunova;  
Tech. Ed.: I.M. Zarkh

PURPOSE: This issue of the Institute's Transactions is intended for meteorological  
and hydrographic specialists working in the field of long-range weather fore-  
casting.

COVERAGE: This collection of articles deals with aspects of extended weather  
forecasting. Individual articles discuss: synoptic conditions of wind  
regimes most favorable to shipping along the Northern Sea Route [Soviet Arctic  
Seas]; synoptic conditions underlying a continuous ice cover in various parts

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Problems in Long-Range Forecasting

SOV/3031

of the Sea of Azov; a method for compiling daily schematic 500-mb contour maps ( $AT_{500}$ ) for 3 days by utilizing an equation of the conservation of vortex velocity and temperature regime; a method for the advance computation of the baric field for periods of 24, 48, and 72 hours; the determination of definite relationships for forecasting air temperature for a natural synoptic period. The results of actual tests in a series of investigations in extended forecasting are cited. References accompany each article.

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Problems in Long-Range Forecasting

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Turketti, Z.I., and O.M. Yakusheva. Computing Prognostic Pressure Fields for  
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Investigations in Compiling Extended Weather Forecasts

100

AVAILABLE: Library of Congress

Card 3/3

TM/jb  
12-19-59

TSEPKANOVA, Ye.I.; KHESINA, B.G.

Estimating the preceding development of atmospheric processes  
and distribution of weather elements in the preparation of  
monthly weather forecasts. Trudy TSIP no.71:44-47 '58.

(MIRA 11:12)

(Weather forecasting)

KHESINA, B.G.

Investigating synoptic climatological relationships of atmospheric processes and temperature anomalies in September. Trudy TSIP no. 92:89-112 '60. (MIRA 14:2)  
(Atmospheric temperature) (Weather forecasting)

KHESINA, B.G.

Forecasting reliability of basic atmospheric processes for  
September. TRUDY TSIP no.115:72-79 '62. (MIRA 16g6)

(Weather forecasting)

"APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5

KHESINA, B.G.

Recurrence of December processes in September. TRUDY TSIP  
no.115:133-140 '62. (MIRA 16:6)

(Meteorology)

APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5"

"APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5

BORISOVA, L.G.; KHESINA, B.G.

Effect of solar activity on the formation of synoptic processes.  
Trudy TSIP no.124:28-32 '63. (MIRA 16:8)  
(Barents Sea—Cyclones) (Kara Sea—Cyclones) (Solar energy)

APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5"

BORISOVA, L.G., kand. geograf. nauk; KHESINA, B.G.

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Weather forecasting for the U.S.S.R. in September 1964.  
Meteor. i gidrol. no.8-61-64 Ag '64 (MIRA 17:8)

APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5"

SHTABOVA, A.I.; KHESINA, B.G., mladshiy nauchnyy sotrudnik

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no.3:61-64 Mr '65. (MIRA 13:2)

1. TSentral'nyy institut prognozov. 2. Glavnnyy inzh.-sinoptik  
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"APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5

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APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5"

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Case of an atypical course of tuberculous meningitis. Vrach.delo  
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(MENINGES—TUBERCULOSIS)

KHESINA, R. L., Cand Med Sci (diss) -- "The course of the acclimatization period among tuberculosis patients arriving on the southern shore of the Crimea from the central and southern regions of the USSR". Moscow, 1957. 22 pp  
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KHESINA, R.L.

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acclimatization to warm climate, thermo-regulation skin tests (Rus))  
(CLIMATE,

acclimatization in pulm. tuberc. to warm climate, thermo-regulation skin tests (Rus))  
(BODY TEMPERATURE,

thermo-regulation skin tests in warm acclimatization in pulm. tuberc. (Rus))

KHESINA, R.L.

*R. L. Хесина* защитила 27/1 1960 г. в Совете Московского государственного научно-исследовательского института курортологии и физиотерапии диссертацию на тему «Течение периода акклиматизации у больных туберкулезом, приезжающих на Южный берег Крыма из центральных и южных районов СССР».

Разработаны показания и противопоказания для больных туберкулезом, приезжающих на Южный берег Крыма. На основании клинических и других методов исследований сердечно-сосудистой, первичной и других систем конкретизирован период акклиматизации этих больных с указанием ряда мероприятий, способствующих улучшению приспособительных реакций организма в этих условиях.

Candidate of Medical Sciences

Dissertations approved by the Higher Attestation Commission in January and February of 1961. Terap. arkh. no. 67: 117-121 '61

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Surgical Clinic, Sanitation-Hygience Faculty, First Moscow  
Branch of the Lenin Med. Inst., (-1944-)

"Chapman's method and its clinical significance for determi-  
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nauk; VOLKOV, P.M., kand.ekon.nauk; VOZNESENSKIY, L.A., nauchnyy  
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KHESTANOV, G.T., podpolkovnik meditsinskoy sluzhby; MAL'TSEV, A.I., podpolkovnik meditsinskoy sluzhby; CHERNOV, I.G., podpolkovnik meditsinskoy sluzhby

Compound treatment of chronic gastritis at the Novye Senzhary Sanatorium.  
Voen.-med. zhur. no.6:78 Je '61. (MIR. 14:8)

(STOMACH—INFLAMMATION)  
(NOVYE SENZHARY—MINERAL WATERS)

GAL'PERIN, F.M.; DEMIN, V.F.; SMIRNOV, A.A.; KHESTANOV, R.Kh.

Nuclear magnetic resonance in nickel. Izv. AN SSSR. Ser. fiz.  
27 no.12:1458-1459 D '63. (MIRA 17:1)

MURAV'IEV, N.V.; KHESTANOVA, L.I.; SHAPOSHNIKOVA, V.V.

Method for analyzing accidents in rural areas. Zdrav. Ros. Feder.  
4 no.12:11-14 D '60. (MIRA 13:12)

1. Iz Respublikanskoy bol'nitsy Severo-Osetinskoy ASSR (glavnyy  
vrach S.S.Khanayev).  
(KIROV DISTRICT (OSSETIA)—ACCIDENTS)

MURAV'YEV, N.V.; Khestanova, L.I. (Ordzhonikidze)

Efforts of the North Ossetian Republic Hospital to improve the  
qualifications of medical workers of the rural area. Zdrav.Ros.  
Feder. 7 no.3:32-35 Mr '63. (MIRA 16'3)  
(OSSETIA, NORTH—PUBLIC HEALTH)  
(OSSETIA, NORTH—MEDICINE—STUDY AND TEACHING)

KHETAGURI, I.

Relying upon the creative initiative of the masses. Sov. profsoiuzy  
16 no.22:19-22 N '60. (MIRA 14:1)

1. Sekretar' Gruzinskogo respublikanskogo soveta profsoyuzov.  
(Georgia—Socialist competition)  
(Georgia—Trade unions)

KHETAGUROV, A.D.

Reserve alkalinity and true urinary acidity in acute myocardial infarction caused by thrombosis of the coronary artery.  
Terap.arkh. 33 no.10:33-38 '61. (MIRA 15:1)

1. Iz kliniki fakul'tetskoy terapii pediatriceskogo fakul'teta  
(dir. - prof. M.I. Zolotova-Kostomarova) II Moskovskogo meditsinskogo instituta imeni N.I. Pirogova.  
(HEART--INFARCTION) (CORONARY VESSELS--DISEASES)  
(ACID BASE EQUILIBRIUM)

BEREZOVSKAYA, Ye. K.; KHETAGUROV, A. D.

Pulmonary adenomatosis. Terap. arkh. 34 no.5:96-98 '62.  
(MIRA 15:6)

1. Iz kafedry fakul'tetskoy terapii (zav. - prof. M. I. Zolotova-Kostomarova) pediatriceskogo fakul'teta II Moskovskogo meditsinskogo instituta imeni N. I. Pirogova i patologoanatomiceskogo otdeleniya 1-y gorodskoy klinicheskoy bol'nitsy (glavnyy vrach - zasluzhennyy vrach RSFSR L. D. Chernyshev)

(LUNGS—TUMORS)

KHETAGUROV, A.D.

Changes in the calcium and inorganic phosphorus of the blood  
in patients with acute myocardial infarct developing after  
coronary thrombosis. Terap. arkh. 35 no.5:15-20 My'63  
(MIRA 16:12)

1. Iz kliniki fakul'tetskoy terapii (dir. - prof. M.I.Zolotova-Kostomarova) pediatriceskogo fakul'teta II Moskovskogo meditsinskogo instituta.

KHETAGUROV, A.I.

KHETAGUROV, A.I.; GUDZHEDZHIANI, B.I.

Work practices using hydraulic filling in Tkibuli Coal Trust mines.  
Trudy Inst. met. i gor. dela AN Gruz. SSR 2:197-215 '49. (MIRA 11:1)  
(Tkibuli--Hydraulic mining)

KHETAGUROV, G., general-polkovnik

Let's teach our sergeants and rely upon them. Komm.Vooruzh.  
Sil 2 no.12:21-28 Je '62. (MIRA 15:8)

1. Komanduyushchiy Severnoy gruppoj voysk.  
(Russia—Army—Noncommissioned officers)

KHETAGUROV, G.A., inzh.

Some shortcomings of a good machine. Transp.stroi. 14  
no.12:56-57 D '64. (MIRA 19:1)

KHETAGUROV, G.D.; SHESTAKOV, V.A.

Utilization of large scale ore mining systems and ways of improving them. Vest. AN Kazakh. SSR 11 no.7:27-38 Jl '54. (MLRA 7:11)  
(Mining engineering)

KHETAGUROV, G.D.; DOBROSERDOV, Ye.I.; YERGALIYEV, A.Ye.; VOLKOV,  
F.I.

Practice of applying high productive systems of mining in  
certain mines. Vest. AN Kazakh. SSR 11 no.9:80-91 S '54.  
(Mining engineering) (MIRA 8:2)

XHETAGUROV, G.D.

Mining systems for thick ore deposits of Dzhezkazgan. Trudy Alt.  
GMMI no.2:91-99 '55.  
(Dzhezkazgan--Mines and mineral deposits) (Mining engineering)

KHETAGUROV, G. D.

137-58-4-6401

Translation from: Referativnyy zhurnal, Metallurgiya, 1958, Nr 4, p 10. (USSR)

AUTHORS: Khetagurov, G. D., Krutikov, P. M.

TITLE: The Effect of the Working Out of a Vein on Flotation Criteria  
(Vliyanie razubozhivaniya na pokazateli flotatsii)

PERIODICAL: Sb. tr. Vses. n.-i. in-ta tsvetn. met., 1956, Nr 1, pp 30-37

ABSTRACT: Laboratory tests were employed to determine the effect of the working out (W) of a deposit on the flotation indices of sulfide ore at one of the fields of a polymetallic source. Pb-Zn-Cu ore, the initial content of which had been 5 percent Pb, 10 percent Zn, and 0.6 percent Cu, was becoming diluted with gangue. The degree of W varied from 0 to 90 percent. The final mixture contained 0.5 percent Pb, 1 percent Zn, and 0.06 percent Cu. It was established that W of the initial ore to contents of about 1.5 percent Pb, 3 percent Zn, and 0.15 percent Cu is accompanied by only a negligible diminution in the extraction of Pb and Zn in the respective froth products. Further W of the ore results in a sharper increase in losses of Pb and Zn in the final tailings. Extraction of Cu undergoes a constant diminution as W increases, and the rate of this decline is considerably more rapid than that

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137-58-4-6401

The Effect of the Working Out of a Vein on Flotation Criteria

in the extraction of Pb and Zn. However, the low Cu content of the initial ore should be taken into consideration. Selective flotation of Pb and Zn remains unchanged for all mixtures and is not dependent upon W.

- A. Sn  
1. Ores--Processes    2. Minerals--Extraction    3. Flotation--Applications

Card 2/2

KHETAGUROV, G.D.; SHESTAKOV, V.A.; BALABOLKIN, A.N.

Basic indexes of the effectiveness of high yield mining systems in certain complex metal ore mines. Trudy Alt. GMNII AN Kazakh. SSR no.3:110-121 '56.

(MLRA 10;2)

(Altai Mountains--Mines and mineral resources)

KHETAGUROV, G.D.

Problems of improving the technology of extracting ore by  
systems of mass breaking-down. Vest.AN Kazakh.SSR 12 no.2:  
20-30 P '56. (Mining engineering) (MIRA 9:6)

*Khetagurov, G.D.*

**KHETAGUROV, G.D.; SHESTAKOV, V.A.**

Determining the maximum ore yield from a block in block caving  
systems. Trudy Alt. GMNII AN Kazakh. SSR 4:52-68 '57. (KIRA 11:1)  
(Mining engineering)

KHETAGIROV, G.D.

Classification of mining systems. Trudy Alt. GMNII A.N Kazakh.  
SSR 6:139-146 '58. (MIRA 12:1)  
(Mining engineering)

KHETAGUROV, G. D.

Evaluating the system of induced level caving. Izv.AN Uz.  
SSR.tekh.nauk no.3:71-83 '61. (MIRA 14:6)

1. Gornyy otdel Akademii nauk UzSSR.  
(Mining engineering)

KHETAGUROV, G.D.

Comparative evaluation of mining systems with induced sublevel  
caving. Trudy Alt. GMNII AN Kazakh. SSR 9:104-118 '60.

(MIRA 14:6)

1. Altayskiy gornometallurgicheskiy nauchno-issledovatel'skiy  
institut AN Kazakhskoy SSR.

(Mining engineering)

KHETAGUROV, G.D.

Improving the system of block ore caving. Izv. AN Uz.SSR.  
Ser.tekh.nauk 6 no.2:65-71 '62. (MIRA 15:7)

1. Gornyy otdel AN UzSSR.  
(Mining engineering)

TROSKAYA, Z.I.; TEMKIN, Z.Ye.; KHETAGUROV, G.D., kand. tekhn. nauk

Quality of nonferrous metal bres and the profitability of  
their production; discussion of the article by B.F. Novozhilov.  
Gor. zhur. no.11:7-11 N '63. (MIRA 17:6)

1. Gosudarstvennyy institut po pioyektirovaniyu predpriyatiy  
tsvetnoy metallurgii, Moskva (for Troskaya, Temkin).
2. Sredneaziatskiy filial Gosudarstvennogo nauchno-issledova-  
tel'skogo instituta tsvetnykh metallov, Almalyk (for Khetagurov).

KHETAGUROV, G.D.

Classification and comparative evaluation of underground  
systems of exploitation of ore beds. Izv. AN Uz. SSR. Ser.  
tekhn. nauk 8 no.2:90-99 '64. (MIRA 17:6)

1. Sredneaziatskiy filial Gosudarstvennogo nauchno-issle-  
dovatel'skogo instituta tsvetnykh metallov.

KHETAGUROV G. I.

PA-17T2

USSR/Medicine - Syphilis  
Medicine - Spirochaeta

Apr 1947

"The Spirochaetocidal Effect in New Methods of Syphilis Treatment in Comparison with the Compact Method of the Author," G. I. Khetagurov. 4 pp

"Vestnik Venerologii i Dermatologii" No 4

From the Leningrad Dermatological and Venereological Institute, Director, S. E. Gorbovitskiy. Some reference to work done by Gormen, Ehrlich, Eltze, and others. In cases of spirochaetosis the usual method of therapy and the droplet method of treatment give results similar to the compact method of Khetagurov. For this reason it is expedient to begin ordinary therapy of syphilis with a large dose (0.6).

17T2

KHETAGUROV, G.I.; RAIK, I.O.

Reinfection with syphilis following penicillin therapy. Vest. vener.  
no.2:45-46 Mar-Apr 1951. (CLML 20:9)

1. Of the Clinic for Male Syphilis, Leningrad Skin-Venereological  
Institute RSFSR (Director--Prof. S.Ye. Gorbovitskiy).

KHETAGUROV, G.I.; RAIK, I.O.

Results of four years of investigation on penacillin therapy of  
syphilis. Sovet. med. no.5:21-23 May 1951. (GIML 20:9)

1. Of Leningrad Skin-Venereological Institute of the Ministry of  
Public Health RSFSR (Director of Institute and Scientific Super-  
visor of Syphilological Department--Prof. S.Ye. Gorbovitskiy).

"APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5

KHETAGUROV, G.S.; YERGALIYEV, A.Ye.; BALOBOLKIN, A.N.; SHESTAKOV, V.A.

Rod-boring in hard rock. Trudy Alt. GUMII AII Kazekh. 1:25-46 154.  
(Boring) (MIRA 10:1)

APPROVED FOR RELEASE: 09/17/2001

CIA-RDP86-00513R000722010004-5"

KHETAGUROV, G.V.

Forms in which silver is found in Kholstinskiy deposit ores  
(Central Caucasus). Izv.vys. ucheb. zav.; tsvet. met. no.3:23-26  
'58.  
(MIRA 11:11)

1. Severokavkazskiy gornometallurgicheskiy institut. Kafedra poleznykh iskopayemykh i poiskovo-razvedochnogo dela.  
(Mizur region--Ore deposits) (Silver ores)

AUTHOR:

Khetagurov, G.V.

SOV/149-58-4-16/26

TITLE:

Mineralogy of the Products of Smelting Gold-bearing  
Ores with Certain Concentrates of East Siberian Origin  
(Mineralogiya produktov plavki zolotosoderzhashchikh  
rud i kontsentratov nekotorykh mestorozhdeniy vostochnoy  
Sibiri)

PERIODICAL: Izvestiya Vysshikh Uchebnykh Zavedeniy, Tsvetnaya  
Metallurgiya, 1958, Nr 4, pp 119-121 (USSR)

ABSTRACT:

Microscopic examination of slags produced in the course  
of the investigation on matte smelting of rich,  
gold-bearing concentrates (described on p.109-118 of  
the present issue of this journal) revealed numerous  
matte inclusions consisting of various sulphides, and  
containing also metallic copper and magnetite. A detailed  
list of all the identified minerals and their  
characteristics is given below:

Bornite ( $Cu_5FeS_4$ ) can be distinguished on freshly polished  
micro-sections by its pink colour and by pale pink and  
bluish films formed rapidly on its surface. It gives a

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Certain Concentrates of East Siberian Origin

positive reaction when etched with  $\text{HNO}_3$ ,  $\text{FeCl}_3$  or  $\text{KCN}$ ,  
is moderately hard and isotropic. It is usually the  
main constituent of the matte inclusions and forms  
often a eutectic with the other sulphides. It contains  
sometimes metallic copper and magnetite (Fig.2).

Chalcopyrite ( $\text{CuFeS}_2$ ) is characterised by yellow colour,  
high reflectivity, positive reaction with  $\text{HNO}_3$  and its  
weak effect on polarised light.

Metallic copper is easily distinguished by its soft  
pink colour, high reflectivity, low hardness and  
positive reaction with all the standard etching reagents  
except  $\text{HCl}$ .

Chalcosine ( $\text{Cu}_2\text{S}$ ) examined by reflected light appears  
pale blue. It can be etched with  $\text{HNO}_3$  and  $\text{FeCl}_3$  and is  
present in its rhombic modification indicated by its  
marked anisotropy.

Cubanite ( $\text{CuFe}_2\text{S}_3$ ) is present in the form of minute,  
allotriomorphic grains characterised by yellowish pink  
colour, high degree of anisotropy and negative reaction

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with HCl, HNO<sub>3</sub> and FeCl<sub>3</sub>.

Magnetite (Fe<sub>3</sub>O<sub>4</sub>) is present in the form of flat particles, no larger than 0.04 - 0.05 mm and is usually found at the matte inclusion/slag interface, or in the top slag layers. Sometimes magnetite can also be present in the interior of the matte inclusions (Fig.2).

Examination of transparent specimens of slag showed that it consisted almost entirely of amorphous silicate glass with a small number of pyroxene (Fig.3), chalcopyrite, arsenopyrite and other sulphide inclusions. The refractive index of the slag was found to be 1.653 which indicated its acid character. The slag contained also some monticellite (CaMgSiO<sub>4</sub>) inclusions. No metallic gold could be observed under

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